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SYNTHESIS O-g-C₃N₄/TiO₂ RUTILE COMPOSITE MATERIAL FOR PHOTOCATALYTIC APPLICATION

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The development of novel materials ensuring the use of solar radiation as an inexhaustible source of renewable and environmentally friendly energy is one of the actual problems of materials science. Scientific research towards of solving this important task showed the expediency of using photocatalytic processes with the participation of semiconductor systems. One of the most well-known catalyst titanium dioxide TiO_2 has photoactivity only in the ultraviolet region of the spectrum that significantly restricts its use. The application of based on undoped graphitelike carbon nitride $g-C_3N_4$ or $g-C_3N_4/TiO_2$ composite catalysts allows using only part of the visible spectrum of solar radiation (with a wavelength of less than 460 nm). It is found that the doping of carbon nitride by oxygen significantly improves its photocatalytic properties to enhancing solar energy utilization. Therefore, to improve the photocatalytic activity of semiconductor photocatalyst, the coupling O-doped $g-C_3N_4$ (O-g-C₃N₄) with rutile TiO₂ is a good strategy. Novel composite material O-g- C_3N_4/TiO_2 was synthesized by gas phase method of deposition of O-doped g- C_3N_4 on particles of rutile powder under the special reactionary conditions of the pyrolysis of melamine. Obtaining O-g-C₃N₄/TiO₂ binary composite was confirmed through various analytical techniques including X-ray powder diffraction (XRD), Fourier transform infrared spectroscopy (FTIR), X-ray photoelectron spectroscopy (XPS) and ultraviolet-visible diffuse reflectance spectra (UV-Vis-DRS) methods. It is found that the absorption spectra of the O-g-C₃N₄/TiO₂ powders show a bathochromic shift of the long-wavelength edge of the fundamental absorption band (to 600 nm) relative to the absorption band of $g-C_3N_4/TiO_2$ (~ 460 nm). As a result, $O-g-C_3N_4/TiO_2$ photosensitivity is observed in the significant part of the visible region and the band gap of synthesized product is determined to be less than 2.4 eV versus 2.7 eV for undoped $g-C_3N_4$ or $g-C_3N_4/TiO_2$. One stage constructing heterojunction structure of O-g- C_3N_4/TiO_2 composite may be used as a low-cost way to avoid the limitations of each component and realize a synergic effect in promoting the efficient generation and separation of charge carriers, thus boosting the photocatalytic activity to enhancing solar energy utilization.

Keywords: O-g-C₃N₄/TiO₂ composite, O-doped carbon nitride, rutile, photocatalyst, melamine, pyrolysis

INTRODUCTION

Using sunlight by semiconductors for the production of renewable and clean energy and purification of water and air from toxic organic pollutants and pathogenic microflora is considered as a promising approach to address the imperative energy and environment problems [1–5]. By harvesting solar light to drive a series of important chemical reactions, semiconductor photocatalysis has emerged as one of the most wastewater fascinating technologies for treatment and converting low-density solar energy into high-density chemical energy.

Titanium dioxide (TiO₂), as one of the most efficient metal oxide semiconductor photocatalyst, is widely used for many applications such as hydrogen evolution, CO_2 reduction and wastewater treatment because of its low cost, high chemical stability [1–4]. However, its photocatalytic activity is limited by rapid photogenerated electron-hole recombination and, most importantly, its wide band gap energy of *ca*. 3 eV (\sim 3.2 and \sim 3.0 eV for anatase and rutile, respectively) which is activated only under ultraviolet (UV) light irradiation that accounts for about 5 % of solar light.

On the other hand, graphite-like carbon nitride $g-C_3N_4$, characterized as a non-toxic and chemically highly resistant material, has been shown to have huge potential as a promising photocatalyst capable of absorbing visible light for renewable energy (in particular, hydrogen production by photocatalytic water splitting) and for broad range of environmental applications (decomposition of organic pollutants and the destruction of pathogenic microflora). However, the application of based on pristine $g-C_3N_4$ catalysts allows us using only part of the visible

spectrum of solar radiation (with a wavelength of less than 460 nm). Thus, in order to further enhance the light harvesting capability of g-C₃N₄, various bandgap engineering strategies, including doping by metal [6, 7] or non-metal atoms, in particular, oxygen [5, 8–12], are used. A number of researches have shown that the doping g-C₃N₄ by oxygen to form O-g-C₃N₄ improved significantly its photocatalytic properties [5, 8–17]. As a result, the absorption spectra of the O-doped carbon nitride powders show a bathochromic shift of the longwavelength edge of the fundamental absorption band (to 498 nm or more) relative to the absorption band of pristine $g-C_3N_4$ (~460 nm) [8, 9, 11, 12]. It is found that O-modified g-C₃N₄ presented intrinsic electronic/band structure modulation [9], resulting in extended light absorption range for more effective visible-light utilization, up-shifted conduction band for stronger reducibility and more effective separation of photogenerated charge carriers, beneficial which are for improving photocatalytic hydrogen evolution activity. For example, according to [9], when using thermally oxidized porous g-C₃N₄, the 1430.1 μ mol \cdot g⁻¹ \cdot h⁻¹ average photocatalytic hydrogen evolution is achieved in 8 h under visible-light irradiation, which was 4.3 times as high as that of the pristine g-C₃N₄ sample (334.3 μ mol \cdot g⁻¹ \cdot h⁻¹). In report [10] is presented that O-doped material (O-g-C₃N₄) shows 6.1 and 3.1 times higher hydrogen evolution reaction activity (with apparent quantum efficiency of 7.8 % at 420 nm) than bulk and even 3D porous g-C₃N₄. to investigation [13], According the photocatalytic hydrogen evolution rate of O-doped g-C₃N₄ was 13.9 times higher than that of bulk g-C₃N₄.

Oxygen doped carbon nitride was successfully applied for production hydrogen peroxide (H₂O₂) which is widely used in wastewater treatment, pulp bleaching, pharmaceuticals etc. [12]. According to [12], oxygen doping g-C₃N₄ changes the catalyst leading properties, to the promoted photocatalytic H₂O₂ production capability in the absence of hole scavengers. As-prepared oxygen doped g-C₃N₄ displays the H₂O₂ concentration of $3.8 \text{ mmol}\cdot\text{L}^{-1}$, more than 7.6 times higher than that of neat $g-C_3N_4$ [12].

Doping graphitic carbon nitride $(g-C_3N_4)$ with oxygen contributes to the effectiveness of

the degradation of pollutants [14]. Thus, ozone treated carbon nitride increase photocatalytic rhodamine B (RhB) degradation constants by approximately 6 times [11], accelerates the photodegradation of methylene blue by a factor of 5 times and 2 times accelerates the generation of H₂ in comparison with untreated graphitic carbon nitride [4]. The 40 % metal-free oxygen doped porous graphitic carbon nitride (O-g-C₃N₄) catalyst can degrade bisphenol A $(15 \text{ mg} \cdot \text{L}^{-1})$ in 240 min with a mineralization rate as high as 56 % [15].

However, the known methods for oxygendoped carbon nitride (O-g-C₃N₄) producing suggest a two-stage process, since the synthesis involves post-treatment of the pre-synthesized undoped g-C₃N₄ with ozone, nitric acid or hydrogen peroxide [8, 13, 15–17]. For example, long-term (10 h) heat treatment process (in a Teflon sealed autoclave at 140 °C) of the presynthesized g-C₃N₄ by H₂O₂ solution was used to obtain oxidized carbon nitride [8]. The pyrolysis method developed in Frantsevich Institute for Problems of Materials Science of NASU (IPM NASU) allows us to obtain oxygen-doped carbon nitride O-g-C₃N₄ in one stage from melamine [18], urea [19-21] or cyanuric acid and urea mixture [22], and O-doped lownitrogen containing carbon nitride O-C₃N from pyridine [23, 24].

Besides the method of carbon nitride modification by oxygen, to increase the activity of the photocatalyst is also promising a different approach: creating g-C₃N₄ based binary photocatalytic composite materials. For efficient utilization, fabrication light the and photocatalytic applications of $g-C_3N_4/TiO_2$ composite materials have attracted much attention [1-4]. The synergistic effect of the combination of g-C₃N₄ and TiO₂ leads to a significant improvement in the absorption of visible light and the effective separation of photo-generated electron-hole pairs. This greatly enhances the photocatalytic activity of g-C₃N₄/TiO₂ composite material, for example, for inactivating against bacteria in water under visible light.

However, both pristine $g-C_3N_4$ and $g-C_3N_4/TiO_2$ composite exhibits photoactivity in the visible spectrum with limited utilization of solar energy (in the part of the visible diapason with wavelength below 460 nm only). While for modified by oxygen graphitic carbon nitride

samples (O-g-C₃N₄) synthesized in IPM [18–22] photosensitivity extends to a much larger region of the visible spectrum. It is found [8–11, 13, 15] that the doping of carbon nitride by oxygen significantly improves its photocatalytic properties to enhancing solar energy utilization. Thereby, to improve the photocatalytic activity of semiconductor photocatalyst, the coupling O-doped graphitic carbon nitride $(O-g-C_3N_4)$ with rutile TiO_2 is a good strategy. Therefore, the of the present was to obtain a aim nanocomposite based on O-doped carbon nitride and rutile phase of titanium dioxide (binary composite O-g-C₃N₄/TiO₂ rutile phase).

EXPERIMENTAL

New composite material $- O-g-C_3N_4/TiO_2$ (rutile phase) was synthesized in accordance with the one-step method developed in IPM for the synthesis of oxygen-doped carbon nitride (O-g-C₃N₄) by gas phase method under the special reactionary conditions of the pyrolysis [25] of melamine [18], urea [19–21] or cyanuric acid and urea mixture [22]. Formation of O-g-C₃N₄/TiO₂ composite was carried out under ambient pressure and pyrolysis-generated selfsupporting atmosphere at the presence of a fixed volume of air. Nanoparticles of O-doped carbon nitride are formed in an vapor-gas reactionary space and are located by means of the deposition and the condensation on particles of rutile powder in more lower temperature (concerning the most highly temperature reaction zone) zones of reactionary space, far from a place of melamine precursor localization. The light yellow colored O-g-C₃N₄/TiO₂ (rutile) was obtained by melamine heating with the rate not exceeding 10 deg·min⁻¹ and heat treatment at 550–580 °C for 1–1.5 h.

Deposition of O-g-C₃N₄ (~ 6 % O) on the particles of rutile powder is confirmed through various analytical techniques including X-ray powder diffraction (XRD), Fourier transform infrared spectroscopy (FTIR), X-ray photoelectron spectroscopy (XPS) and ultraviolet-visible diffuse reflectance spectra (UV-Vis-DRS) methods.

The powder X-ray diffraction (XRD) measurements were obtained from 5 to 80 ° of 20 by a DRON-UM diffractometer with CuK_{α} -radiation ($\lambda = 0.154$ nm) and nickel filter.

Fourier transform infrared spectroscopy (FTIR) in the reflectance mode was determined

between $450-4000 \text{ cm}^{-1}$ with the spectral resolution of 8 cm⁻¹ using a Nexus Nicolet FTIR spectrometer (Thermo Scientific) equipped with a Smart Collector reflectance accessory. Samples under investigation were prepared by pressing with KBr in 1:100 ratio.

X-ray photoelectronic spectra (XPS) of the samples were measured with a photoelectron spectrometer manufactured by SPECS Surface Nano Analysis Company (Germany). The system is equipped with a PHOIBOS 150 hemispherical analyzer. A base pressure of a sublimation ion-pumped chamber of the system was less than 5×10^{-10} mbar during the present The experiments. $Mg K_{\alpha}$ radiation (E = 1253.6 eV) was used as a source of XPS spectra excitation. The XPS spectra were measured at the constant pass energy of 25 eV. The energy scale of the spectrometer was calibrated by setting the measured Au4 $f_{7/2}$ and Cu2 $p_{3/2}$ binding energies to 84.00±0.05 and 932.66 \pm 0.05 eV, respectively, with regard to $E_{\rm F}$. For investigated sample, all the spectral features are attributed to the constituent element corelevels or Auger lines.

Diffuse reflectance spectra (DRS) of the powders were recorded in spectral diapason 200–1000 nm using a Perkin-Elmer Lambda Bio 35 UV-Vis with an integrating sphere Labsphere RSA-PR-20 with BaSO₄ as a scattering standard and converted into ultraviolet–visible (UV-Vis) absorption spectra by the Kubelka-Munk function [26].

RESULTS AND DISCUSSION

At the heat treatment of melamine at 570 °C for 1.5 h in the most highly temperature reaction zone, in the placed in lower temperature zone the ceramic crucible with rutile powder, the light yellow colored powdery product are formed. The product obtained at 570 °C for 1 h has an even lighter shade. The obtained samples were O-g-C₃N₄/TiO₂-1.5h denoted as and $O-g-C_3N_4/TiO_2-1h$. To confirm the accordance of the obtained light yellow powder with the $O-g-C_3N_4/TiO_2$ composition, in the experiments the synthesized (IPM) O-doped carbon nitride $(O-g-C_3N_4)$ [18], pristing $g-C_3N_4$ and TiO_2 (rutile phase) powdered samples also used. To understand the formation of O-g-C₃N₄/TiO₂, the crystal structure of O-g-C₃N₄, pristine g-C₃N₄, rutile phase TiO₂ and presumably obtained $O-g-C_3N_4/TiO_2$ composite were firstly investigated. Diffractograms of products O-g-C₃N₄/TiO₂-1.5h and O-g-C₃N₄/TiO₂-1h are practically identical. In the diffractograms of the both rutile titanium dioxide (Fig. 1, curve 4) and obtained light yellow powdered product O-g-C₃N₄/TiO₂-1.5h (Fig. 1, curve 3) there are strong peaks characteristic for the rutile phase (JCPDS No. 21-1276) with the most intense reflexes at $2\theta = 27.4^{\circ}$ (110), 36.2° (101) and 54.4 ° (211). As is known, there are two characteristic reflexes at $2\theta = 27.49 \circ (002)$ and 12.40 ° (100) in XRD patterns of pristine $g-C_3N_4$ (Fig. 1, curve 1). These reflexes are caused by an interlayer stacking of aromatic heteroatomic rings with an interplane distance (0.324 nm) and the periodicity of stacking of heptazine fragments in one layer (0.714 nm). However, for O-g-C₃N₄/TiO₂ rutile composite sample (Fig. 1,

curve 3), one typical peak of both-g-C₃N₄ and $O-g-C_3N_4$ (002) is not clearly observed due to the coverage by that of rutile TiO_2 (110), while the absence of another peak at 12.4 ° of $O-g-C_3N_4$ (Fig. 1, curve 2) is owing to its high dispersion on the surface of the composites. Occurrence of an additional reflex (wide galo) at $2\theta = 21.45$ ° (d = 0.414 nm) in the diffraction pattern of O-doped carbon nitride (O-g-C₃N₄) (Fig. 1, curve 2) is caused, as it was offered in [18, 19], with partial distortion of a planarity of its polymeric network $((C_6N_7)-N)_n$ because of a dearomatization of some heterocycles at an oxidation. It is important to note that in the diffractogram of the O-g-C₃N₄/TiO₂ binary composite a wide halo at 21.45 ° (characteristic for O-doped carbon nitride) is also clearly observed (Fig. 1, curve 3).



Fig. 1. X-ray diffraction spectra of: *l* – g-C₃N₄, *2* – O-g-C₃N₄, *3* – composite O-g-C₃N₄/TiO₂-1.5h, *4* – rutile TiO₂ samples

To stronger evidence the existence of O-g-C₃N₄ and TiO₂ in the as-obtained sample, the surface chemistry of O-g-C₃N₄/TiO₂-1.5h was analyzed by XPS analysis. Four elements (carbon, nitrogen, oxygen and titanium) can be identified in the O-g-C₃N₄/TiO₂ composite sample in the survey scan XPS spectra (Fig. 2 *a*).

Under CVD conditions of the melamine precursor excess, the surface of the particles of rutile powder is almost completely covered with O-modified carbon nitride (according to XPS, the content of TiO₂ on the surface of the binary material is less than 10 %). This fact explains the weak intensity of the Ti peak on the survey spectrum. As one can see from the Fig. 2 b, the XPS Ti2*p* core-level spectrum recorded for this sample corresponds to that of titanium dioxide. As shown in the high-resolution Ti2*p* spectrum (Fig. 2 *b*), peaks at 458.8 and 464.5 eV are assigned to Ti2 $p_{3/2}$, and Ti2 $p_{1/2}$ of TiO₂.

In Fig. 3 *a*, the C1*s* spectra of O-g-C₃N₄/TiO₂-1.5h could be deconvoluted into two main peaks cored at 284.6 and 288.0 eV. The major C1*s* peak at ~ 288 eV is identified as the tertiary carbon C–(N)₃ in the g-C₃N₄ lattice. The weaker C1*s* peak at ~ 284.6 eV (which always present in the spectra of carbon nitride) is assigned to adventitious carbon species in good agreement with literature data [8, 16, 27]. Furthermore, the third weak peak at higher binding energy of

~290 eV was detected in O-g-C₃N₄/TiO₂-1.5h sample, which was accredited to C=O or C–O [8, 16, 17], suggesting the partial oxidation and formation of oxygen functional groups.

The deconvolution of the N1s spectrum (Fig. 3 b) of the O-g-C₃N₄/TiO₂ composite

14000 N 1s а 12000 10000 C KI I Intensity (a.u) O 1s 8000 N KLL C_{1s} O KLL Ti 2p 6000 4000 2000 0 400 1000 800 600 200 Λ Binding energy (eV) Ti 2p_{3/2} Ti 2p b 5000 459.3 4800 Intensity (a.u) 4600 Ti 2p_{1/2} 465.3 4400 4200 4000 3800 475 470 465 460 455 450 445 Binding energy (eV)

Fig. 2. XPS survey (a) and Ti2p core-level spectrum (b) for the composite O-g-C₃N₄/TiO₂-1.5h

The deconvolution of the O1s spectrum of the O-g-C₃N₄/TiO₂ composite showed the presence of two main maxima (Fig. 4). These peaks at binding energies about 531.5 and 529.4 eV in the high-resolution O1s XPS spectrum can be assigned to oxygen atoms in Odoped carbon nitride (C=O, C-O) and rutile (Ti-O), respectively. It is important to note that in the binary composite sample (O-g-C₃N₄/TiO₂) the O1s signal intensity of oxygen-containing groups (C=O, C-O) is significantly stronger than that of Ti-O. (In accordance with XPS results, the content of oxygen bound to carbon in the O-g-C₃N₄/TiO₂-1.5h sample is 5.6 %).

Further demonstration of the formation of O-g-C₃N₄ on rutile TiO₂ nanopowder surface is given by FTIR spectra (Fig. 5). IR spectra of all the samples (Fig. 5) contain, first of all, the peak near 810 cm⁻¹, which could be ascribed to the characteristic out-of-plane bending mode of tri-*s*-triazine (heptazine) units in both g-C₃N₄ and O-g-C₃N₄ [18, 19]. Besides, in the IR spectra of

showed the presence of two main peaks centered at 398.5 and 399.8 eV, which were attributed to sp^2 -hybridized nitrogen (C=N-C) and tertiary sp^3 -hybridized nitrogen (N-(C)₃) atoms in the carbon nitride network, respectively. all samples there are several distinct intensive absorption bands in the 1200–1650 $\rm cm^{-1}$ region, which also are characteristic for g-C_3N_4 and

 $O-g-C_3N_4$ [18, 19] and correspond to the stretching vibrations of aromatic CN bonds in condensed nitrogen-carbon heterocycles.



Fig. 3. High-resolution XPS C1s (a) and N1s (b) core-level spectra for the composite O-g-C₃N₄/TiO₂-1.5h

It should be noted that the IR spectrum of a sample $O-g-C_3N_4/TiO_2-1h$ (Fig. 5, curve 2), which obtained with a shorter heat treatment time (1 h), is weakly structured. As one can see from the Fig. 5, curve 3, in the IR spectrum of obtained with a longer thermal treatment (for 1.5 h) sample $O-g-C_3N_4/TiO_2-1.5h$ (which contains more than 5% oxygen) there are absorption bands characteristic of heptazine fragments of the g-C₃N₄ structure, as well as bands of the -OH, >C=O and -COOH groups, which are characteristic only for oxidized carbon nitride. The peak at 1089 cm⁻¹ is attributed to the stretching vibration of C-O, together with the

N–H and O–H signals around 3000–3600 cm⁻¹ generally related to the presence of hydroxyl groups. Most intensive signal becomes to emerge at ~ 1750 cm⁻¹, which suggests the formation of carbonyl (carboxyl) groups. The presence of carboxyl groups is an evidence of a stretching band at ~ 2700 cm^{-1} , characteristic for bond –O–H of carboxyl group. Note that the presence oxygen-containing functional groups which confirms the oxygen is doped into O-g-C₃N₄ lattice, is in good agreement with the literature data [8, 12, 14, 16, 28]. However, some researchers believe (and provides convincing evidence also) that O doping preferentially occurs at two-coordinated N position [10, 13, 15]. So, in [10] by experiment and DFT computation were identified that oxygen atoms preferentially replaced nitrogen atoms at two-coordinated N position. According to [10], after

 Ar^+ etching to remove surface layer, O-g-C₃N₄ sample still shows this lattice oxygen, confirming that O atoms are uniformly doped in the matrix.



Fig. 4. High-resolution XPS O1s core-level spectrum for the composite O-g-C₃N₄/TiO₂-1.5h



Fig. 5. IR spectra of: $l - g-C_3N_4$, $2 - \text{composite O-g-C_3N_4/TiO_2-1h}$, $3 - \text{composite O-g-C_3N_4/TiO_2-1.5h}$, $4 - \text{O-g-C_3N_4}$ samples

In attempt to obtain more evidence relative to the structure alternation due to the O doping, in [15] EPR was conducted to detect the spin state of unpaired electrons. The bulk g-C₃N₄ exhibits one single Lorentzian line with a *g* factor of 2.003 due to the unpaired electron of the π -bonded aromatic rings [15]. It has been found [15] that in the EPR spectrum of O-doped carbon nitride a shoulder peak appears, probably because substituting the sp^2 N atom in the heptazine unit with an O atom creates one unpaired electron. Thus, oxygen position in O-modified carbon nitride network currently still under discussion. Probably in the structure of the O-g-C₃N₄ lattice, both the addition of oxygen in the form of functional groups and the substitution of two-coordinated nitrogen with oxygen is possible. So, according to [13], the FTIR spectrum and XPS analysis of O1s and C1s verifies that doping O atoms either take the place of two-coordinated nitrogen atoms or exist as hydroxide radicals linked to C atoms in the C=N-C ring.

For the O-g-C₃N₄/TiO₂ samples, despite the strong light scattering of TiO_2 at low wavenumbers as well as the fact that he surface of the particles of rutile powder is almost completely covered with O-doped carbon nitride

(according to XPS), a weak wide signal near 800 cm^{-1} was observed (Fig. 5, curves 2, 3).

UV-vis diffuse reflectance spectra (DRS) were used to study the light absorption properties of pure O-g-C₃N₄ and O-g-C₃N₄/TiO₂ rutile composite compared to pristine rutile phase TiO₂ and undoped g-C₃N₄ (Fig. 6 *a*).



Fig. 6. UV-Vis diffuse reflectance spectra of the powders converted by the Kubelka-Munk function (*a*) and estimation of band gap energies E_g (*b*) of: $1 - g-C_3N_4$, $2 - O-g-C_3N_4$, $3 - \text{composite O-g-C_3N_4/TiO_2-1h}$, $4 - \text{composite O-g-C_3N_4/TiO_2-1.5h}$, $5 - \text{pure TiO_2 samples}$

The optical absorption spectra reveal that the pure undoped $g-C_3N_4$ and rutile phase TiO_2 samples feature an intrinsic semiconductor-like absorption in the blue region of the visible spectrum and ultraviolet region respectively. There is a bathochromic shift of the absorption edge which expands to the significant part of the

visible light region for all synthesized powders doped with oxygen (from 460 nm for undoped g-C₃N₄ to 600 nm with absorption trailing extended to more than 700 nm for O-g-C₃N₄ in g-C₃N₄ \rightarrow O-g-C₃N₄/TiO₂-1h \rightarrow O-g-C₃N₄/TiO₂-1.5h \rightarrow O-g-C₃N₄ range). It is assumed that in comparison with bulk g-C₃N₄, the absorption band edge of O-g-C₃N₄ is red shifted due to the presence of O-doping, and samples with more O-dopants show more obvious red shift [10]. The absorption band edge of O-g-C₃N₄/TiO₂ composite is located between pure TiO₂ and oxygen modified O-C₃N₄ samples, which further confirmed the electronic coupling of these two components in the O-g-C₃N₄/TiO₂ heterojunction. Overall, the absorption intensity of all oxygen-doped samples (pure $O-g-C_3N_4$, $O-g-C_3N_4/TiO_2-1h$ and especially O-g- $C_3N_4/TiO_2-1.5h$) is significant higher than that of pristine g-C₃N₄ in the full spectrum and the absorption edges of oxygenated products are red shifted. It is assumed [13], that the red shift in the absorption wavelength indicates that the introduction of oxygen results in the absorption of more light energy to produce more electron-hole pairs, photogenerated which improvement in contribute to the the photoactivity of the catalysts. The band gap energies of the materials were estimated by the Kubelka-Munk equation by transforming the spectra into $(\alpha hv)^{1/2}$ versus hv (for g-C₃N₄, the *n* is $\frac{1}{2}$ for the indirect band gap semiconductor). The band gap energy (E_g) estimated from the

intercept of the tangents to the plots of $(\alpha hv)^{1/2} vs$ photon energy decreases in the range O-g-C₃N₄ < O-g-C₃N₄/TiO₂-1.5h < O-g-C₃N₄/TiO₂-1h < g-C₃N₄ < TiO₂ (rutile phase) and has values of 2.2, 2.3, 2.4, 2.6, and 3.0 eV, respectively, as shown in the Fig. 6 *b*. The obtained data confirm that the construction of heterojunction structure can be responsible for the overall improvement of light adsorption intensity and show the possibility of practical use of oxygen-doped O-g-C₃N₄/TiO₂ composite in photocatalysis.

CONCLUSIONS

Novel composite material $O-g-C_3N_4/TiO_2$ rutile with photoactivity in significant part of visible light spectral diapason was obtained. Thus, the integration of two favorable characteristics (heteroatom introducing and the construction of heterojunction structure) in the novel composite $O-g-C_3N_4/TiO_2$ (rutile phase) may be used as a cost-effective way to avoid the drawbacks of each component and realize a synergic effect in promoting the efficiently transfer of photo-generated electron-hole pairs to restrict the recombination, thus boosting the photocatalytic activity.

Синтез композитного матеріалу О-g-C₃N₄/TiO₂ рутил для фотокаталітичного застосування

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Створення сучасних матеріалів, що забезпечують використання сонячного випромінювання як невичерпного джерела відновлюваної і екологічно чистої енергії – одне з найбільш актуальних завдань матеріалознавства. Наукові дослідження в напрямку вирішення цього важливого завдання показали доцільність використання фотокаталітичних процесів за участю напівпровідникових систем. Один з найбільш відомих каталізаторів діоксид титану ТіО2 має фотоактивність тільки в ультрафіолетовій області спектра, що істотно обмежує його використання. Застосування каталізатора на основі недопованого графітоподібного нітриду вуглецю $g-C_3N_4$ або композиту $g-C_3N_4/TiO_2$ дозволяє використовувати тільки частину видимого спектру сонячного випромінювання (з довжиною хвилі менше 460 нм). Встановлено, що допування нітриду вуглецю киснем значно покращує його фотокаталітичні властивості. Таким чином, для підвищення фотокаталітичної активності напівпровідникового фотокаталізатора хорошою стратегією є поєднання О-допованого g-C₃N₄ (O-g-C₃N₄) з рутилом TiO₂. Новий композитний матеріал O-g-C₃N₄/TiO₂ був синтезований газофазним методом за допомогою осадження О-допованого g-C₃N₄ на частинкі порошку рутилу в особливих реакційних умовах піролізу меламіну. Отримання бінарного композиту O-g-C₃N₄/TiO₂ було підтверджено різними аналітичними методами, включаючи рентгенівську порошкову дифракцию, інфрачервону спектроскопію з Фур'єперетворенням, рентгенівську фотоелектронну спектроскопію і спектри дифузного відбиття в ультрафіолетовому та видимому діапазонах. Встановлено, що в спектрах поглинання порошків O-g-C₃N₄/TiO₂ спостерігається батохромний зсув довгохвильового краю смуги фундаментального поглинання (до 600 нм) відносно смуги поглинання g-C₃N₄/TiO₂ (~ 460 нм). В результаті фоточутливість O-g-C₃N₄/TiO₂ спостерігається в значної частини видимої області спектра, а ширина забороненої зони синтезованого продукту становить менше 2.4 еВ проти 2.7 еВ для недопованого g-C₃N₄ або g-C₃N₄/TiO₂. Одностадійний метод створення гетероперехідної структури – композиту O-g-C₃N₄/TiO₂ може бути використаний як економічно вигідний спосіб уникнути недоліків кожного компонента і реалізувати синергетичний ефект в сприянні генерації і розподілу носіїв заряду, таким чином посилюючи фотокаталітичну активність матеріалу для більш ефективного використання сонячної енергії.

Ключові слова: композит O-g-C₃N₄/TiO₂, O-допований нітрид вуглецю, рутил, фотокаталізатор, меламін, піроліз

Синтез композитного материала O-g-C₃N₄/TiO₂ рутил для фотокаталитического применения

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Создание современных материалов, обеспечивающих использование солнечного излучения в качестве неисчерпаемого источника возобновляемой и экологически чистой энергии – одна из наиболее актуальных задач материаловедения. Научные исследования в направлении решения этой важной задачи показали целесообразность использования фотокаталитических процессов с участием полупроводниковых систем. Один из наиболее известных катализаторов диоксид титана TiO₂ обладает фотоактивностью только в ультрафиолетовой области спектра, что существенно ограничивает его использование. Применение катализатора на основе недопированного графитоподобного нитрида углерода $g-C_3N_4$ или композита g-C₃N₄/TiO₂ позволяет использовать только часть видимого спектра солнечного излучения (с длиной волны менее 460 нм.) Установлено, что допирование нитрида углерода кислородом значительно улучшает его фотокаталитические свойства. Таким образом, для повышения фотокаталитической активности полупроводникового фотокатализатора хорошей стратегией является сочетание О-допированного $g-C_3N_4$ (O-g-C₃N₄) с рутилом TiO₂. Новый композитный материал O-g-C₃N₄/TiO₂ был синтезирован газофазным методом посредством осаждения О-допированного $g-C_3N_4$ на частицы порошка рутила в особых реакционных условиях пиролиза меламина. Получение бинарного композита O-g-C₃N₄/TiO₂ было подтверждено различными аналитическими методами, включая рентгеновскую порошковую дифракцию, инфракрасную спектроскопию с Фурье-преобразованием, рентгеновскую фотоэлектронную спектроскопию и спектры диффузного отражения в ультрафиолетовом и видимом диапазонах. Установлено, что в спектрах поглощения порошков O-g-C₃N₄/TiO₂ наблюдается батохромный сдвиг длинноволнового края полосы фундаментального поглощения (до 600 нм) относительно полосы поглощения g-C₃N₄/TiO₂ (~ 460 нм). В результате фоточувствительность O-g-C₃N₄/TiO₂ наблюдается в значительной части видимой области спектра, а ширина запрещенной зоны синтезированного продукта составляет менее 2.4 эВ против 2.7 эВ для недопированного g-C₃N₄ или g-C₃N₄/TiO₂. Одностадийный метод создания гетеропереходной структуры — композита O-g-C $_3N_4/TiO_2$ может быть использован как экономически выгодный способ избежать недостатков каждого компонента и реализовать синергетический эффект в содействии генерации и разделению носителей заряда, таким образом усиливая фотокаталитическую активность материала для более эффективного использования солнечной энергии.

Ключевые слова: O-g-C₃N₄/TiO₂, O-допированный нитрид углерода, рутил, фотокатализатор, меламин, пиролиз

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